

IR and AFM studies of diamonds for detector applications



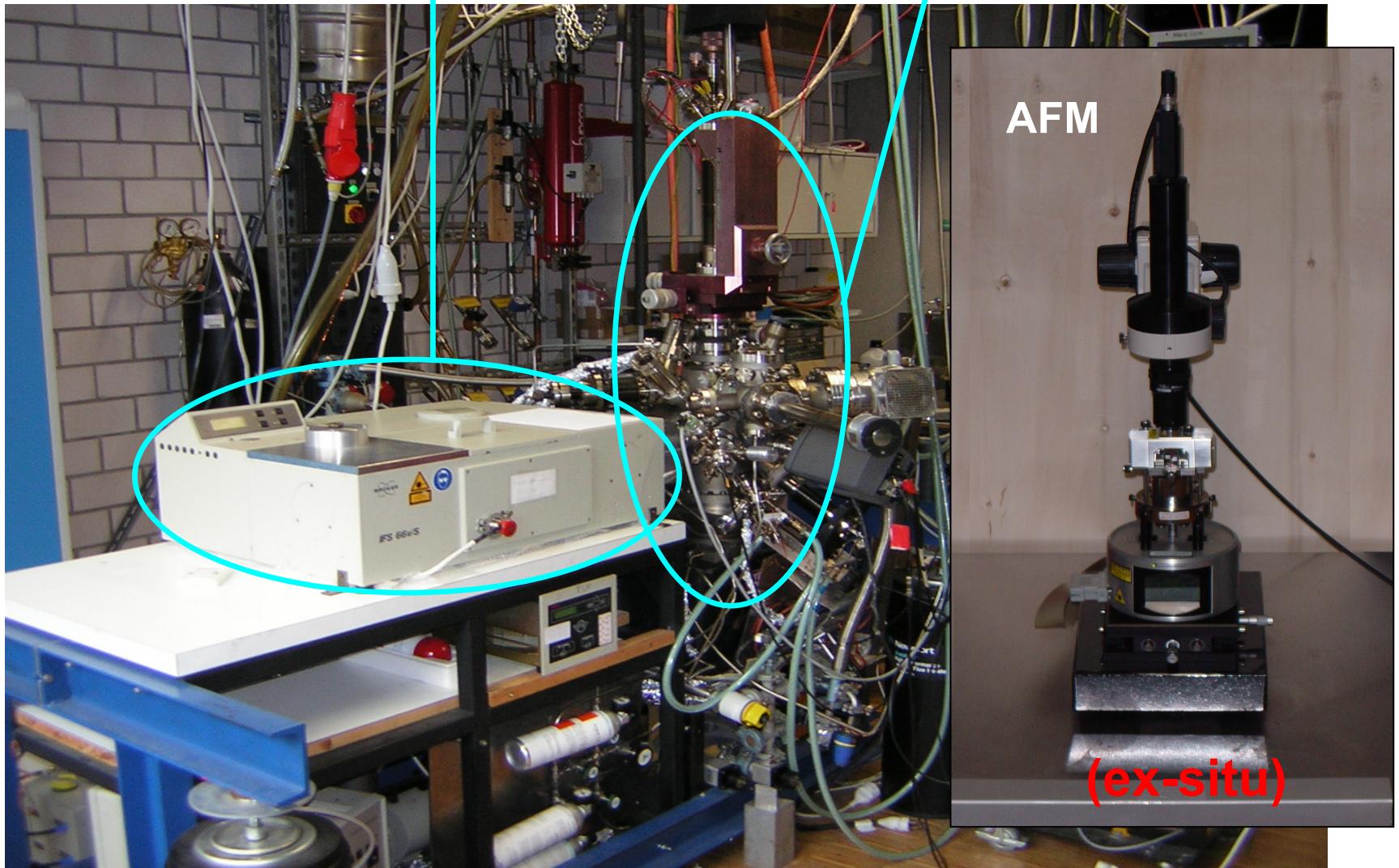
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3. NoRDHia workshop
31.08.06

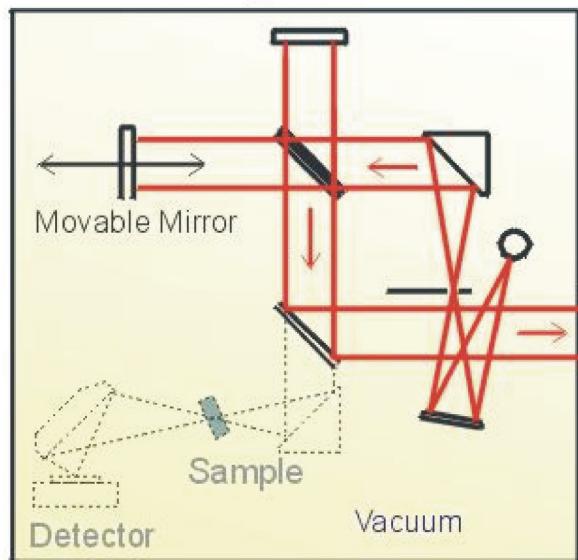
Content

- Experimental setup and what we measure
 - Ultra high vacuum (UHV) chamber
 - Atomic force microscopy (AFM)
 - IR-spectroscopy
 - Drude-model
- The diamond (100) surface
 - surface terminations and related electronic properties
 - Diamond-metal interfaces
 - AFM pictures of E6 diamonds
- First results: Cr on C(100)-(1x1):O
- Outlook

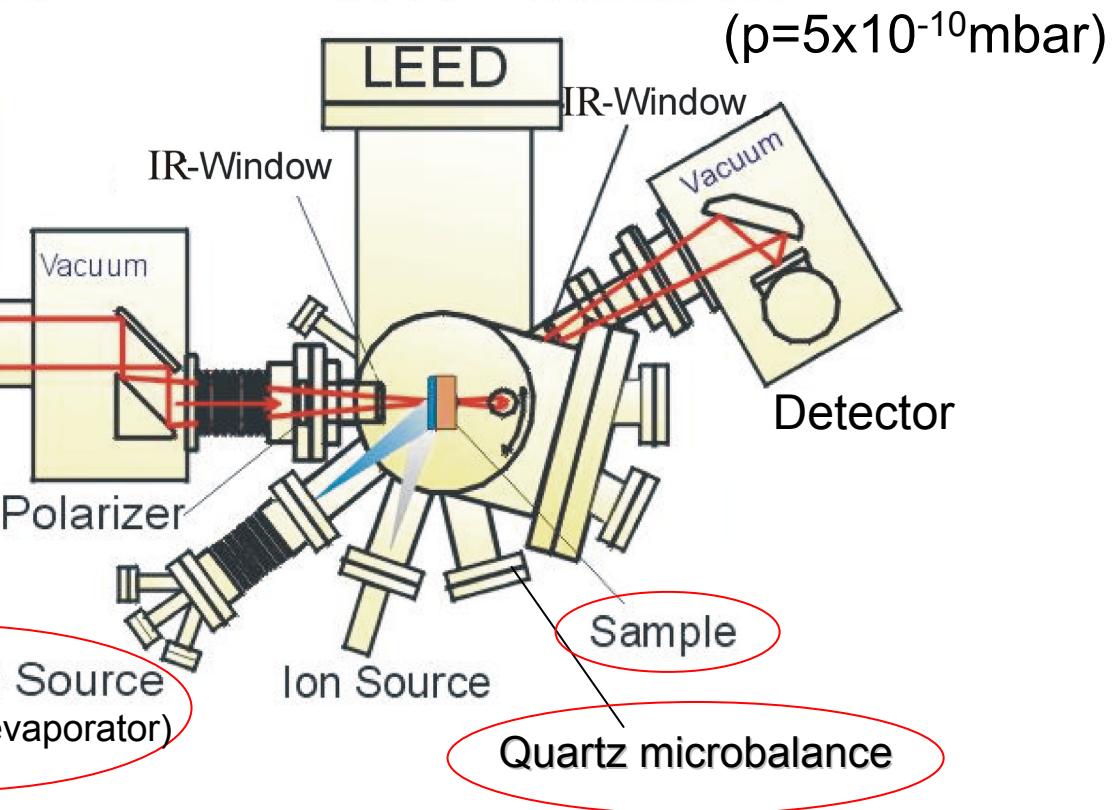
FTIR spectrometer and UHV chamber



FTIR-Spectrometer



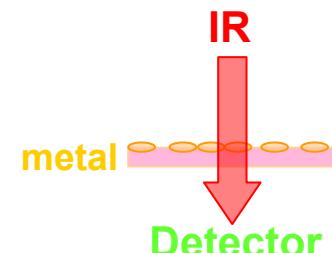
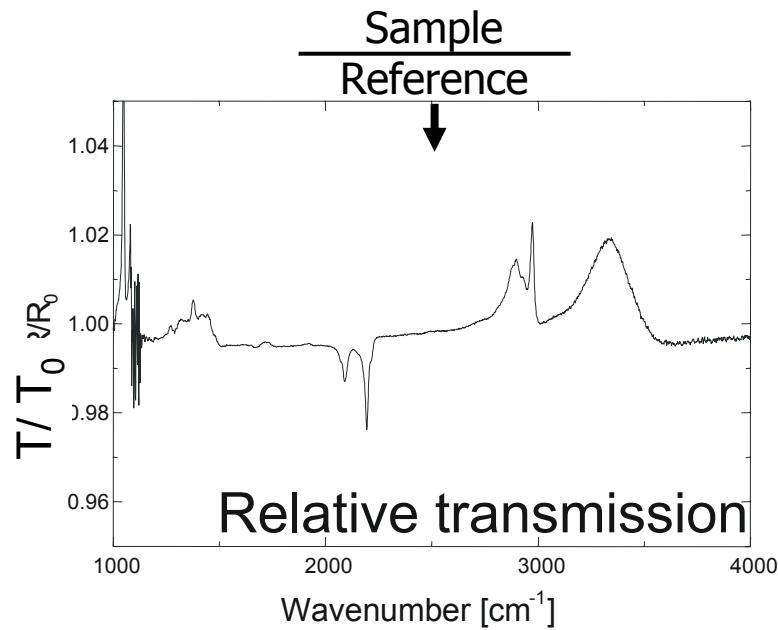
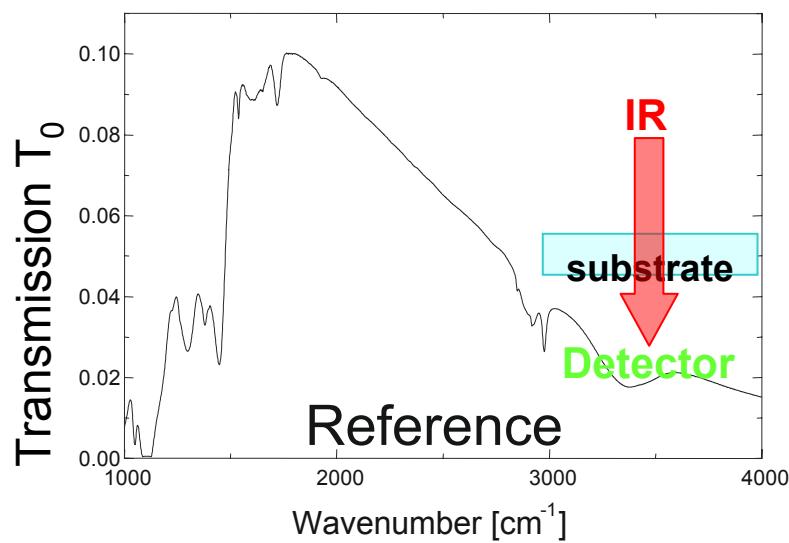
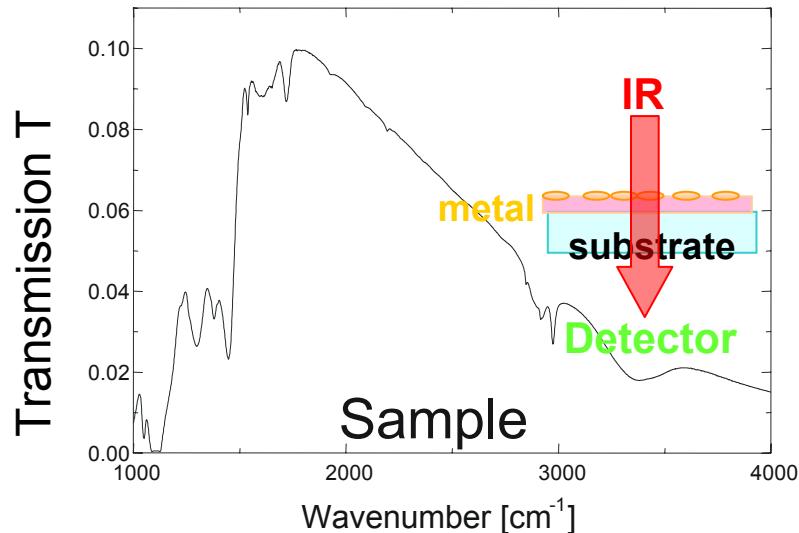
UHV-Chamber



accessible sample temperatures: 40K to 1300K

surface preparation: heating, sputtering

Measurement of relative spectra



wavenumber: $\omega = \lambda^{-1}$

What we measure

- Mostly transmittance at normal incidence
- For ultrathin metal films ($d \ll \lambda / n_{\text{film}}$)

$$\frac{T_{\text{film/substrate}}}{T_{\text{substrate}}} \approx 1 - \frac{2 \cdot d \cdot \omega \cdot \text{Im} \epsilon_{\parallel \text{film}}(\omega)}{c \cdot (1 + n_{\text{substrate}})},$$

$$\epsilon_{\text{film}} - \epsilon_{\infty} = \frac{i\sigma(\omega)}{\omega \epsilon_0}$$

In-plane
conductivity
measurement
without
electrical
contacts

$$\epsilon_1(\omega) = \epsilon_{\infty} - \frac{\omega_p^2}{\omega^2 + \omega_{\tau}^2}$$

$$\epsilon_2(\omega) = \frac{\omega_{\tau} \cdot \omega_p^2}{\omega^3 + \omega \cdot \omega_{\tau}^2}$$

$$\omega_p = \sqrt{\frac{ne^2}{\epsilon_0 m_{\text{eff}}}}, \omega_{\tau} = \frac{1}{\tau}$$

Thin film corrections

$$\omega_\tau \rightarrow \omega_\tau(d) = \omega_\tau^{bulk} + \omega_\tau^{surf}(d)$$

due to scattering of electrons at the surface,
depends strongly on roughness

$$\omega_p \rightarrow \omega_p(d) = \beta(d) \cdot \omega_p^{bulk}$$

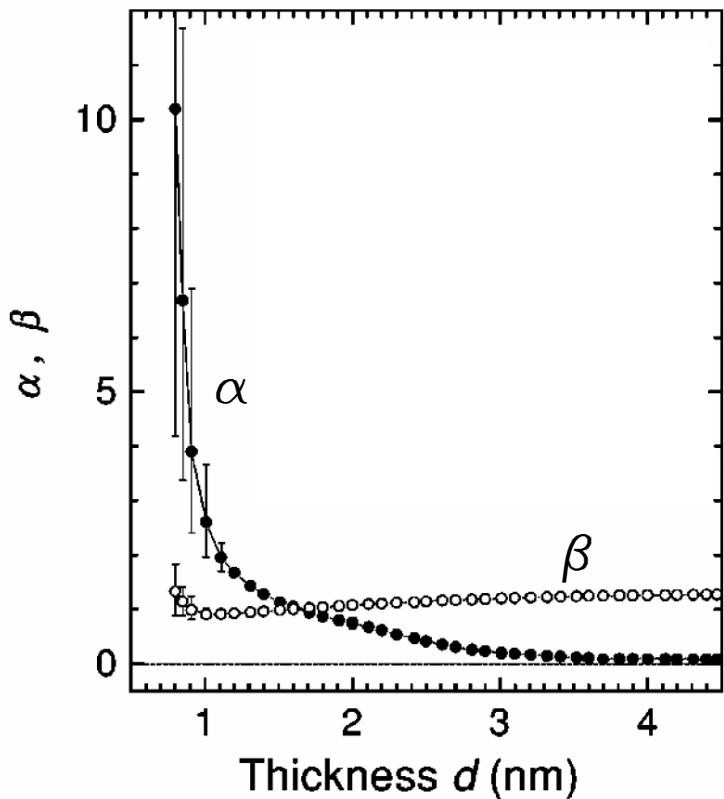
band bending at interfaces causes change of m_{eff} and
Schottky effect at metal - semiconductor contacts leads
to electron transfer which influences n

$$\varepsilon_1(\omega, d) = \varepsilon_\infty - \frac{\omega_p(d)^2}{\omega^2 + \omega_\tau(d)^2}$$



$$\varepsilon_2(\omega, d) = \frac{\omega_\tau(d) \cdot \omega_p(d)^2}{\omega^3 + \omega \cdot \omega_\tau(d)^2}$$

Example: Fe on MgO(001)



Parameters α (solid circles) and β (open circles) for a Fe-film on MgO.
 $\alpha=0$ and $\beta=1$ correspond to bulk values
for the relaxation rate and the plasma
frequency, respectively.
Below a thickness of $\sim 0.8\text{nm}$, the film
cannot be described by a Drude-type
dielectric function.

Fahsold et al., PRB 61, 14108

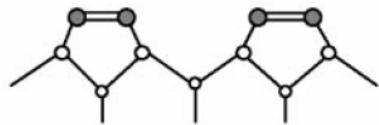
Thin film growth

Metal thin films can be trapped in many equilibrium conditions, depending on several preparation conditions, e. g.

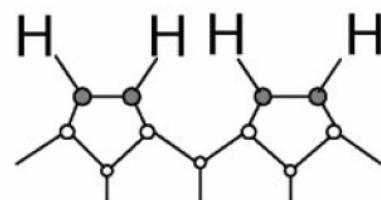
- evaporation rate
- background pressure
- substrate temperature
- substrate structure**
- ...

The diamond (100) surface

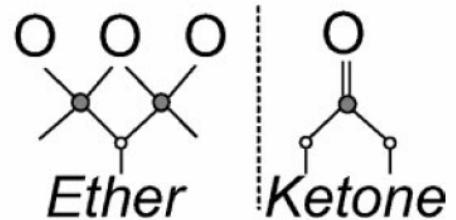
C(100)-(2x1)



C(100)-(2x1):H



C(100)-(1x1):O

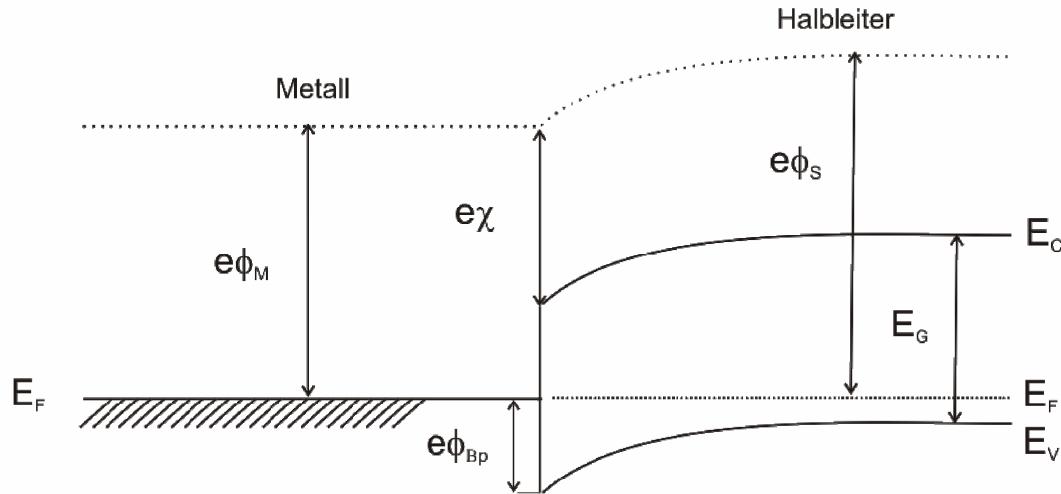


Maier et al., PRB 64, 165411 (2001)

Hydrogen terminated diamond exposed to air exhibits a high surface conductivity and is hence inapplicable for detector applications

Metal-diamond interface

p-Halbleiter



band bending at interface due to constant Fermi energy

barrier height for p-semiconductor
(if no Fermi level pinning occurs):

$$\Phi = E_g + \chi - \Phi_M$$

Electron affinitys of diamond surfaces

	χ [eV]
C(100)-(2x1)	0.5
C(100)-(1x1):O	1.7
C(100)-(2x1):H	1.3

Ways to control the barrier height

- evaporating carbide forming metals (Ti, Mo, Cr) and subsequent annealing ($\sim 600^\circ\text{C}$)
- removing oxidation in UHV before metallization

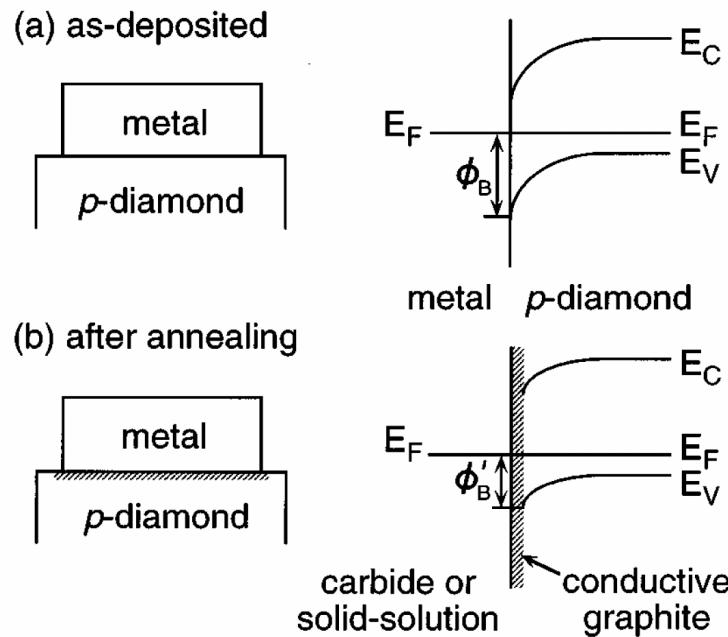


FIG. 7. Interfacial band diagrams at the *p*-diamond/metal interface: (a) as-deposited metal on diamond, (b) conductive graphite layer induced by carbide formation.

Yokoba, J. Appl. Phys. 81, 6815 (1997)

Werner, Semicond. Sci. Technol. 18, 41 (2003)

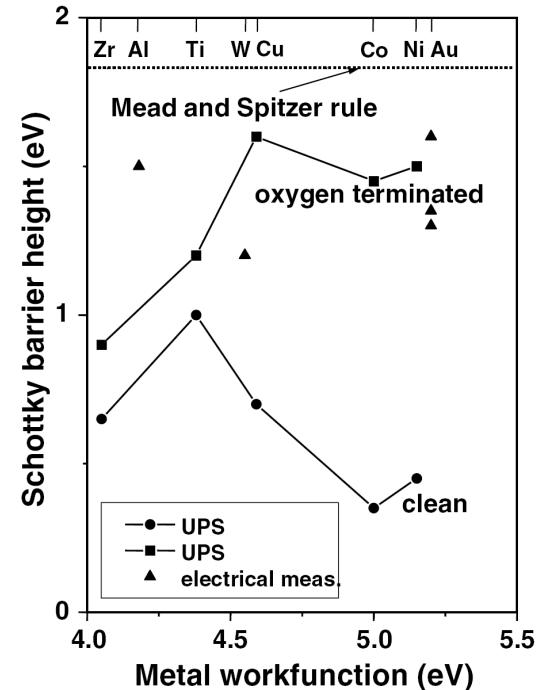


Figure 1. Comparison of the measured Schottky barrier heights on clean and oxygen-terminated surfaces.

Oxidation of diamond surface

- Wet-chemical treatment with H_2SO_4 and KNO_3 at 240°C to oxidize the surface and remove conductive graphite phases
- alternatively, the surface can be treated by oxygen-plasma
- Examination of oxidation quality with IR-spectroscopy? (it is known that acid treatment does leave some residual hydrogen on the surface)

Diffuse reflectance infrared spectroscopy
of diamond crystallites
stretching mode of a carbonyl group
at 1731cm^{-1}

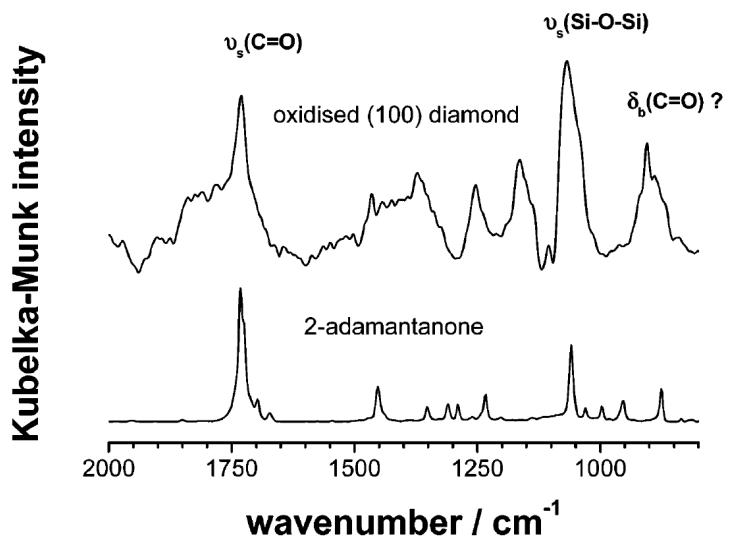
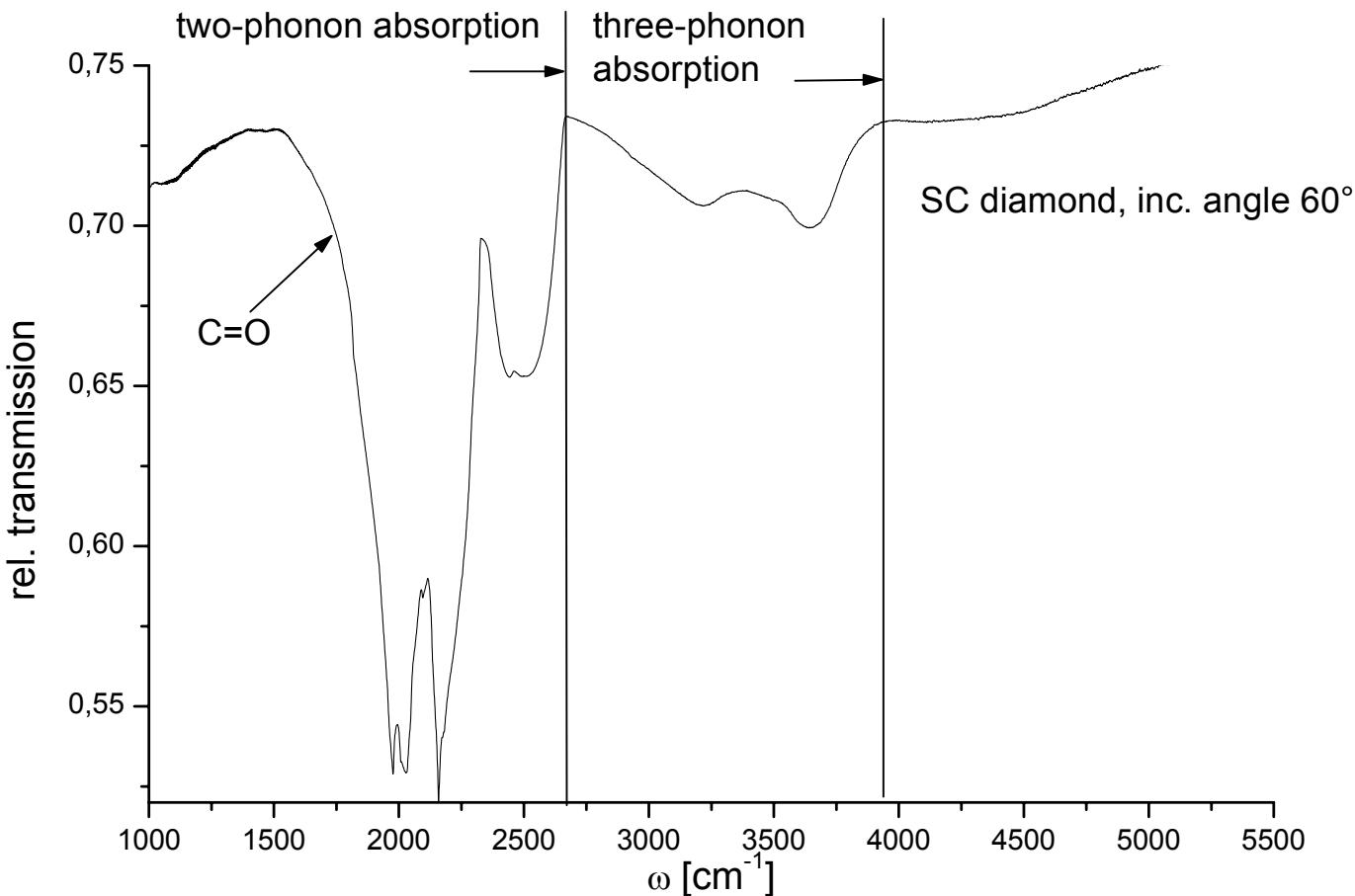


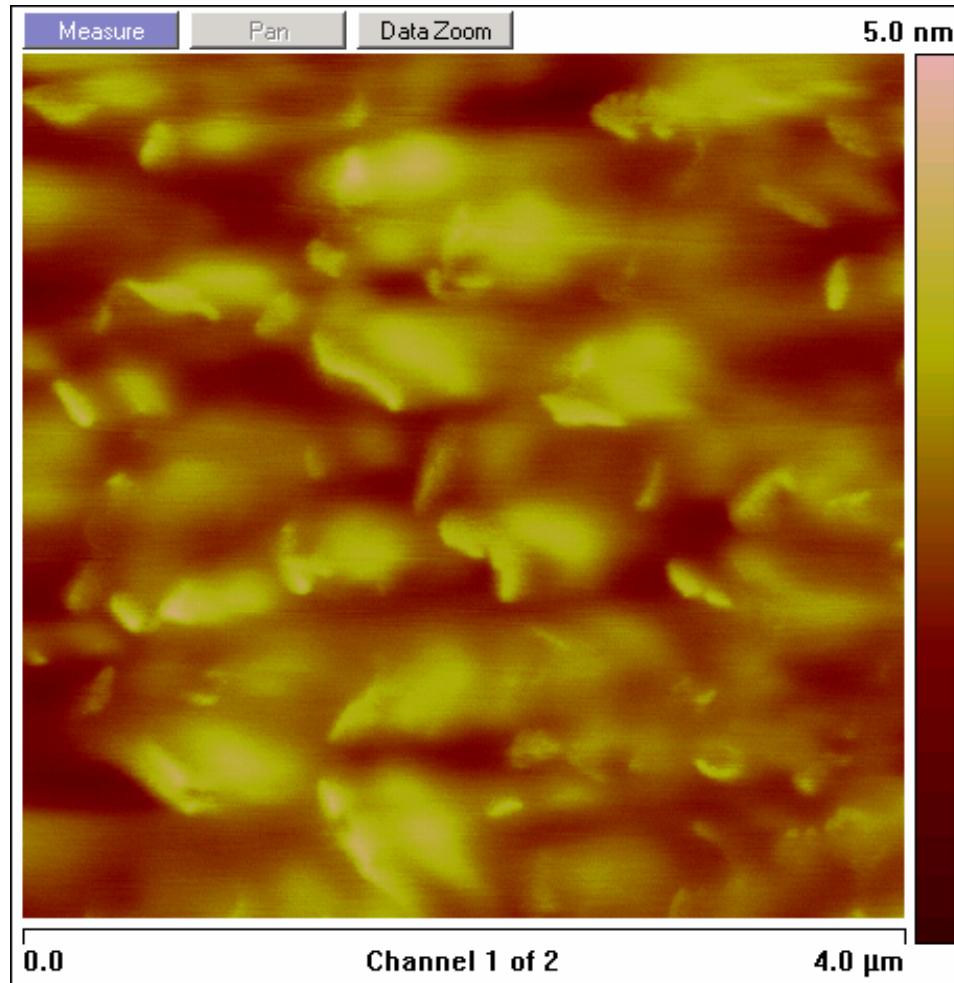
Figure 1. DRIFT spectra of (100) diamond oxidized at 700°C , for 10 min in 150 Torr O_2 , and powdered 2-adamantanone – the vibrational spectrum of 2-adamantanone has been assigned²¹ previously.

IR-transmission spectrum



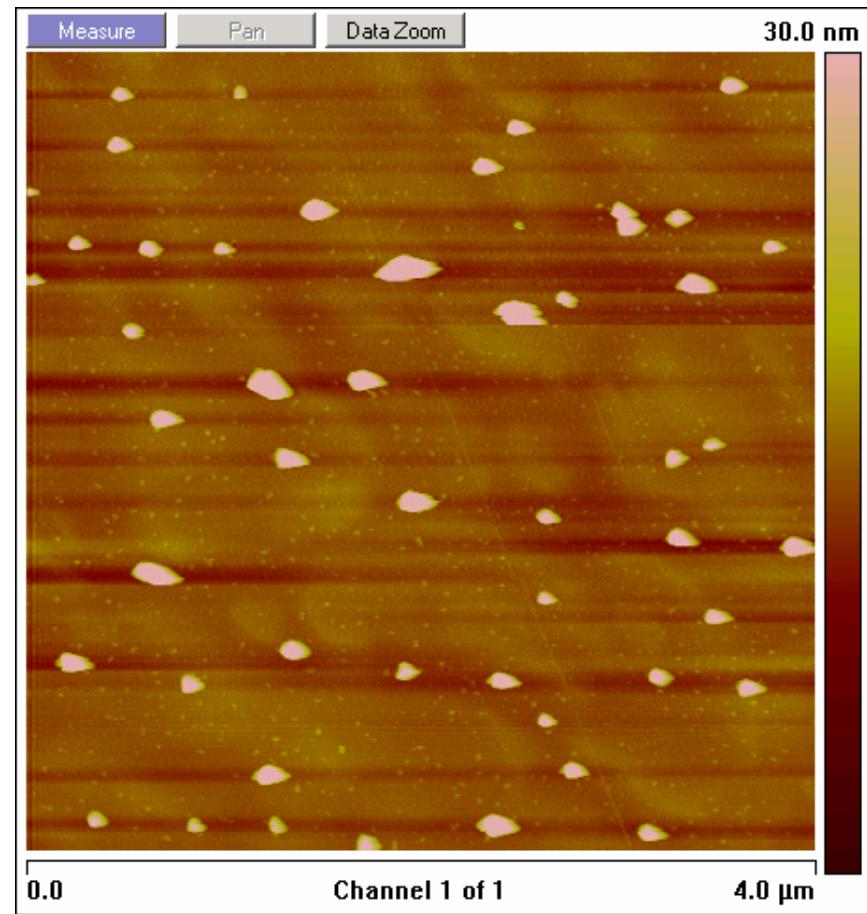
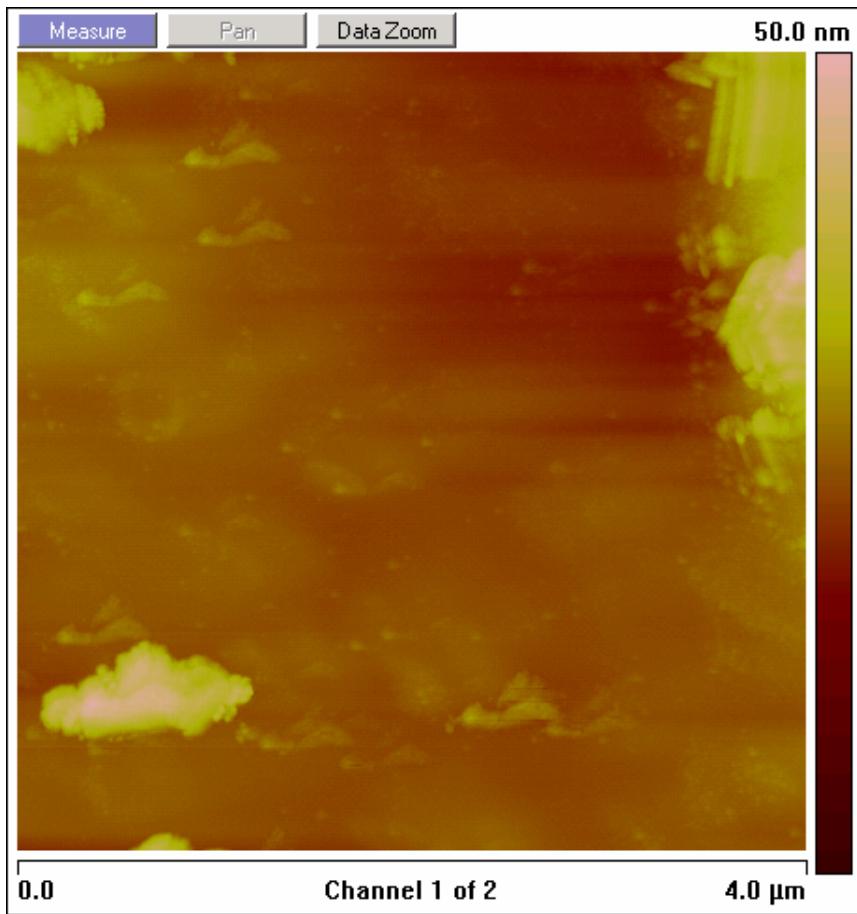
insufficient surface sensitivity in transmission geometry

AFM pictures of an E6 diamond



rather smooth, roughness of a few nanometers

but in other areas...



Different morphology for different places on the diamond.
Influence of wet-chemical treatment?

In order to achieve reliable information about metal film growth, a well defined substrate is needed!

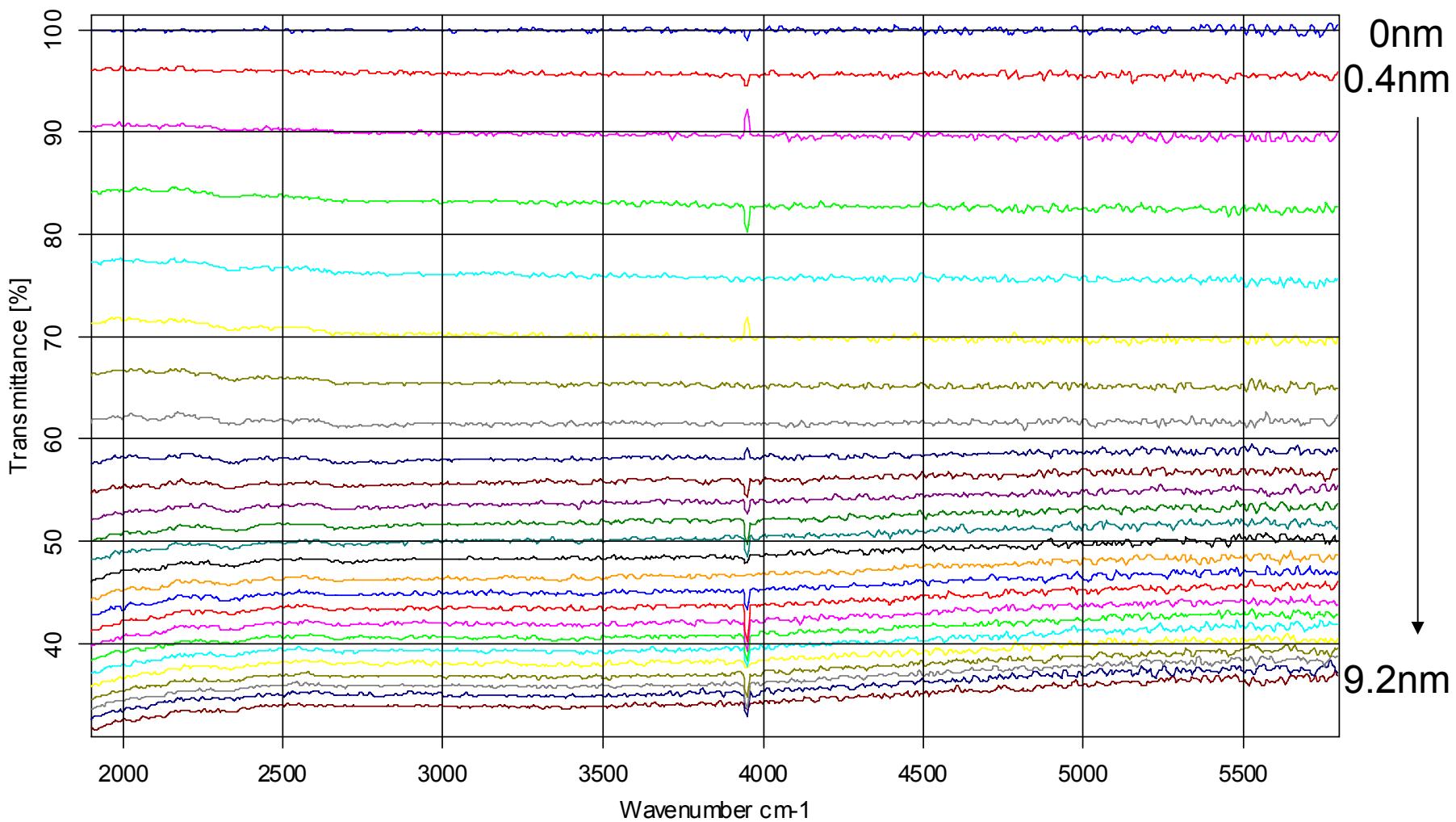
Experimental procedure for Cr-film growth

- oxidizing with H_2SO_4 and KNO_3 at 240°C
- transferring into UHV and heating to 400°C in order to clean surface
- evaporating Cr with a rate of ~900sec/nm, substrate at room temperature



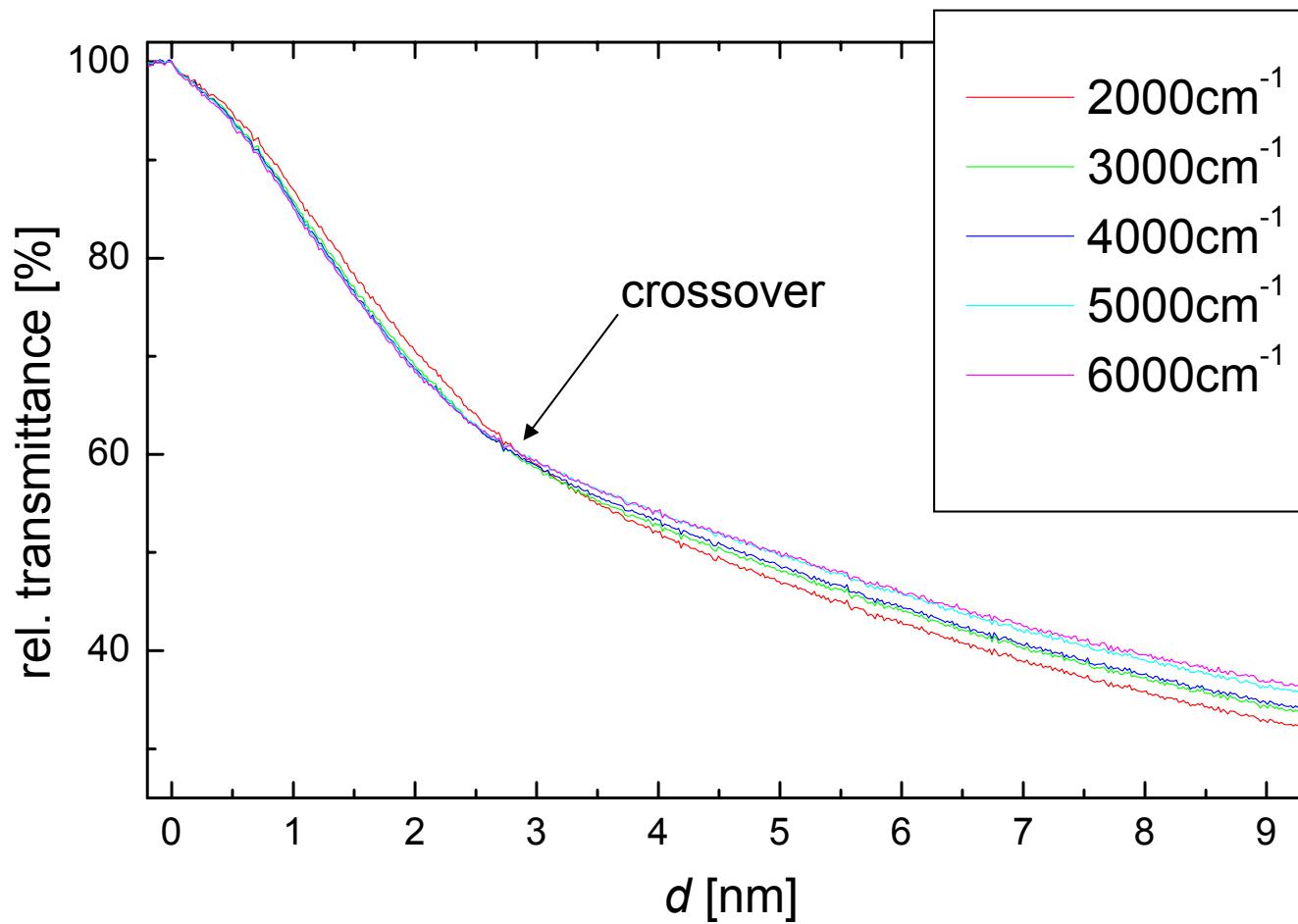
LEED-pattern of C(100)-(1x1):O

First results: Cr on C(100)-(1x1):O

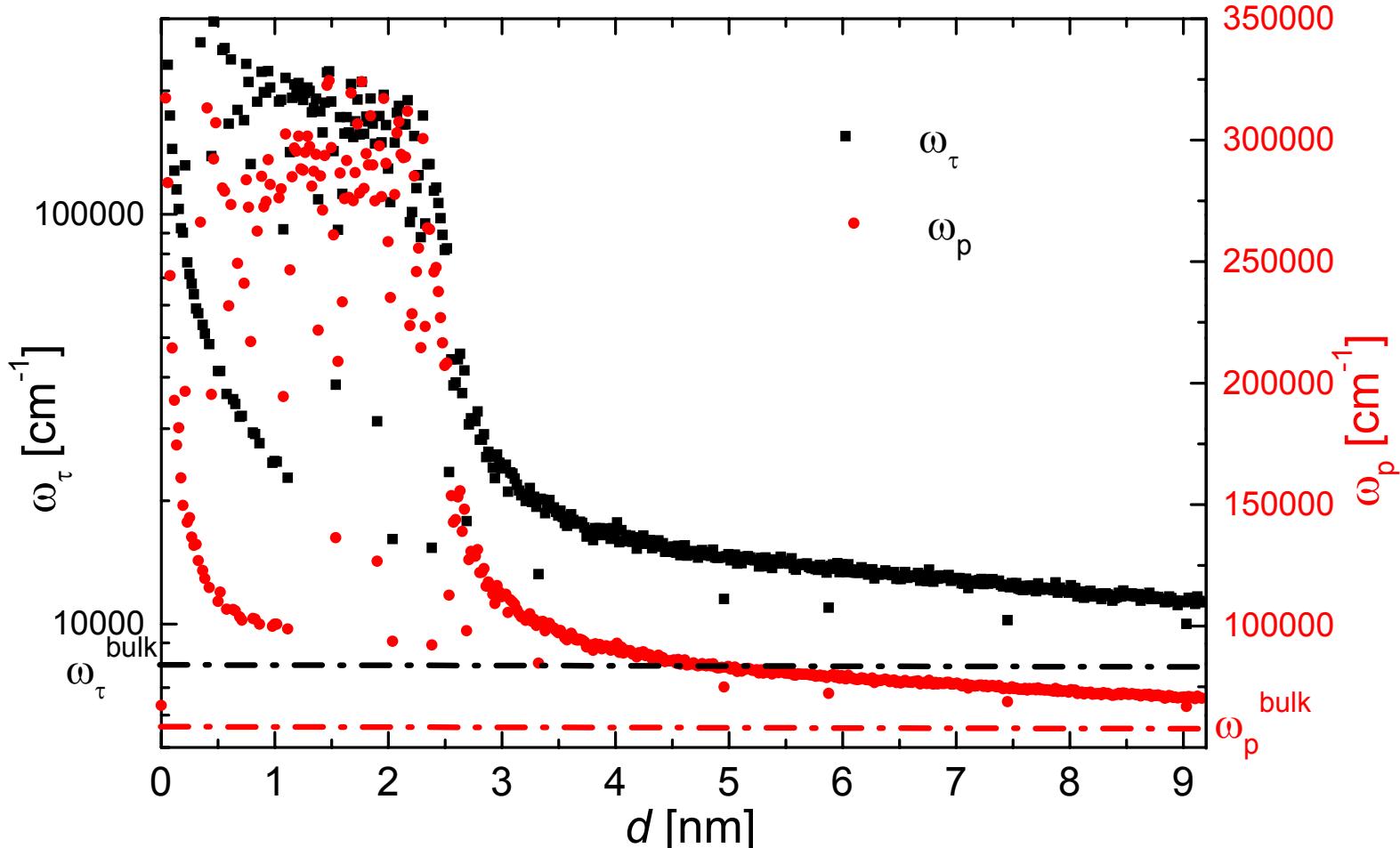


IR-spectra measured during Cr-evaporation

Development of transmission at different frequencies

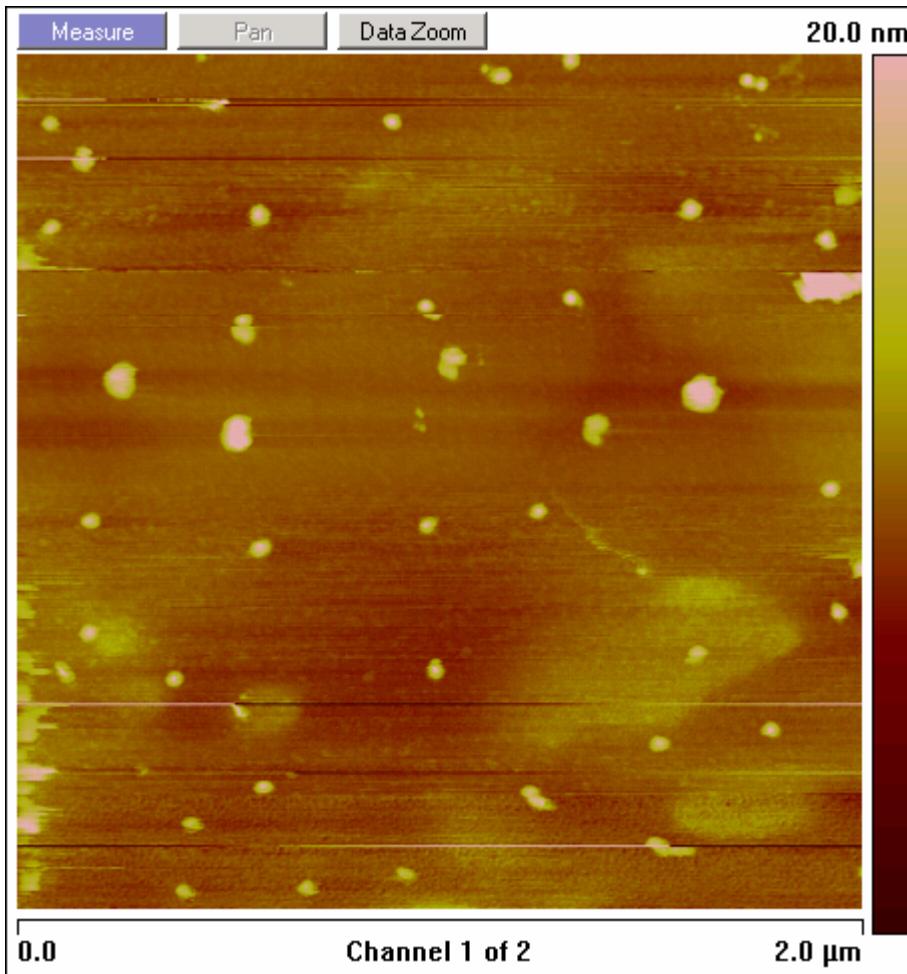


Effective Drude parameters

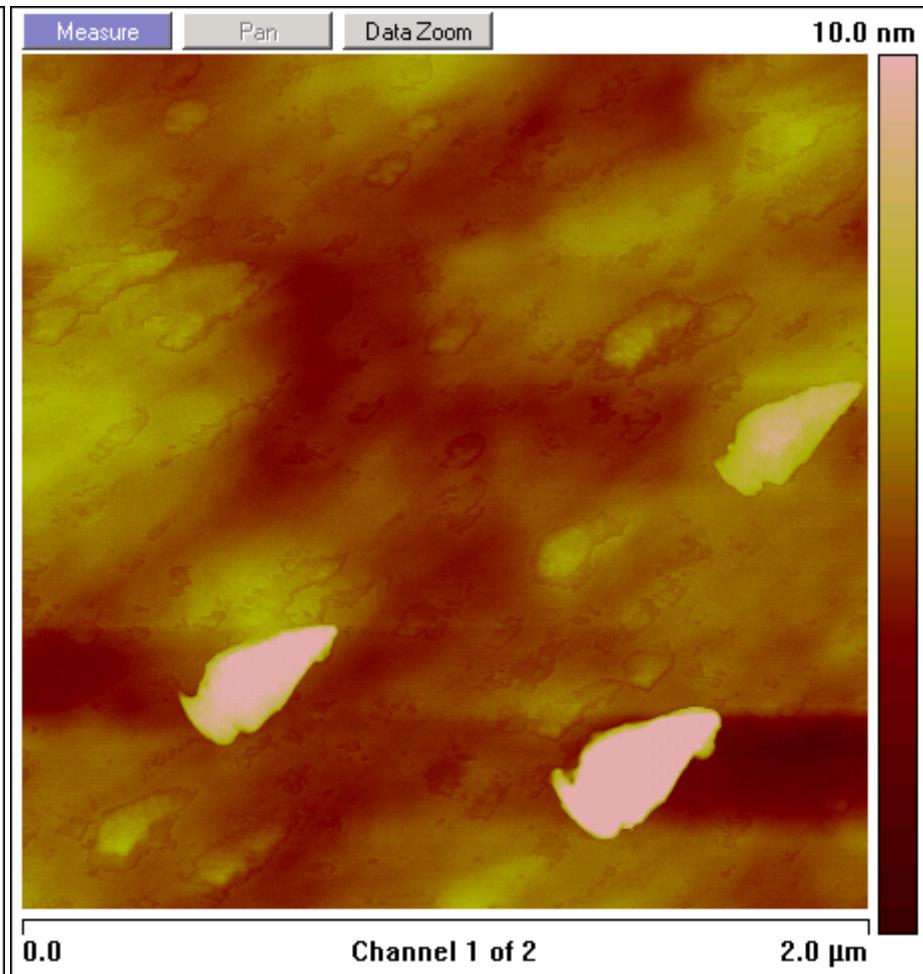


- spectra fitted from 3000 to 5000cm^{-1} , constant ω_p and ω_τ for bulk Cr assumed
- from 2.5 nm on, the IR-spectra of the film can be described by a Drude-type dielectric function

AFM measurements



substrate nearby



Cr-Film

Outlook

- influence of diamond morphology on film growth has to be studied
- comparison of wet-chemical and oxygen-plasma oxidation
- investigation of chromium-carbide formation

Thanks

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